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# Microwave-assisted synthesis and magnetic studies of cobalt oxide nanoparticles

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## 1. Introduction

Over the past few decades, various synthetic methods have been established for the synthesis of transition metal oxide nanoparticles. In particular, the morphology-controlled synthesis of Cobalt (II, III) oxide  $(Co_3O_4)$  with a spinel structure (Fd3m) has been extensively investigated. Several technological applications in the field of heterogeneous catalysts [1], solid state sensors [2], electrochromic sensors [3], anode materials in lithium ion rechargeable batteries [4], energy storage [5] and magnetic materials [6] have made  $Co_3O_4$ the main target of material chemists. Because of the influence of the particle size and morphology on the properties of materials, the controlled preparation of the cobalt oxide nanoparticles has become a necessity for the scientists. Thermal decomposition of cobalt precursors [7], chemical spray pyrolysis [8], chemical vapour deposition [9], sol-gel method [10] and solvothermal synthesis [11] are some of the general methods employed to synthesize  $Co_3O_4$ . However, these methods are either time consuming or require expensive instruments. Microwave-assisted synthetic route is an appropriate approach under such circumstances; the benefits of narrow size distribution and high purity being the added advantages [12].

The size reduction of metal oxides leads to novel properties which makes them potential materials for applications in various fields. It is anticipated that antiferromagnetic nanoparticles exhibit

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## ABSTRACT

An efficient microwave-assisted route has been used to synthesize nanoparticles of cobalt oxide. The particles were well characterized by transmission electron microscopy (TEM) which showed that the average diameter of the particles is around 6 nm. X-ray diffraction (XRD) studies further confirmed the formation of the spinel  $Co_3O_4$ . Purity of the products was detected by Fourier transform infrared spectroscopy (FTIR) combined with thermal gravimetric analysis (TG/DTG). The magnetic measurements revealed a small hysteresis loop at room temperature indicating a weak ferromagnetic nature of the synthesized  $Co_3O_4$  nanoparticles. The magnetic moment of the particles was measured to be 4.27  $\mu_{eff}$ .

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induced permanent magnetic moments due to lack of internal structural perfection and/or uncompensated spins on the surface of the particles [13].  $Co_3O_4$  is one such antiferromagnetic material having Neel temperature of ~30 K. They have gained increased attention for exhibiting quantum tunneling of magnetization [14] and in many other applications involving the magnetic quantum effects [15,16]. No wonder in the recent years, research on synthesis and magnetic studies of  $Co_3O_4$  nanoparticles of desired shape and size has gained momentum.

In the present work, we exploit the benefits of the microwaveassisted synthetic route to prepare uniform  $Co_3O_4$  nanoparticles with narrow size distribution. The results show that the experiment yielded high purity  $Co_3O_4$  showing a ferromagnetic behaviour.

#### 2. Experimental

All the chemicals used were of analytical grade and were used as received without further purification. In a typical synthesis, 2 g  $Co(NO_3)_2 \cdot 6H_2O$  was dissolved in 10 ml ethylene glycol. To this 8 g of surfactant trioctyl phosphine oxide (TOPO) was added and the solution was sonicated for 30 min to attain homogeneity. The homogeneous solution was then transferred to a Teflon lined vessel and subjected to microwave irradiation (QWave 1000, Questron Technologies Corp., Korea) at high power. The Teflon vessel is connected to an exhaust to drain off any vapours produced during the reaction. After 5 min, a violet coloured solution was obtained, which was centrifuged and washed with acetone and dried at 60 °C. The precursor thus obtained was heated at 40 °C for 3 h in air to finally yield the blackish oxide product.

Investigations on the morphology of the Co<sub>3</sub>O<sub>4</sub> nanoparticles and their electron diffractions (ED) were carried out using a transmission electron microscope (TEM, JEOL, 2000FXII) operated at 200 kV. The X-ray powder diffraction analysis was conducted on a JEOL X-ray Diffractometer at a scanning rate of 2°/min with 2 $\theta$  ranging from 20 to 80°, using Cu K<sub>\alpha</sub> radiation ( $\lambda$  = 1.5406 Å). FTIR (THERMO) spectrophotometer has been used to determine the formation of the nanoparticles from the precursor. Thermal analysis was carried out (EXSTAR-6000) from room temperature to 1000 °C at a heating rate of 10°/min under nitrogen atmosphere. The magnetic

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Fig. 1. Bright field TEM image of Co<sub>3</sub>O<sub>4</sub> nanoparticles.

properties were assessed with a Vibration Sample Magnetometer (ADE-DMS EV-7 VSM). Magnetic susceptibility of the samples at room temperature was measured by a Magnetic Susceptibility Balance, Sherwood Scientific, Cambridge.

## 3. Results and discussion

#### 3.1. Transmission electron microscopy (TEM)

The size and shape of the  $Co_3O_4$  nanoparticles were studied by TEM and crystal structure was identified by electron diffraction spectroscopy (EDS). Fig. 1 shows the bright-field TEM images of the sample. The particles show some kind of agglomeration with an average diameter  $D_{\text{TEM}}$  of 6 nm. Sampling about 150 particles from different TEM micrographs showed that they are almost uniform with a standard deviation of <13%. The particle size data are summarized in Fig. 2. The Polynomial and Gaussian fitment curves in respect of the particle size distribution have also been provided in Fig. 2. As can be seen from the figure, the particle distribution mode is 5 nm and the size of the particles are controlled mostly within 4-8 nm. It is also evident from the figure that more than 67% of the particles are in the range of 5–7 nm and no particle is <2 nm. The orientation of the nanoparticles is random, which is evidenced from the SAED patterns (Fig. 3). The ring patterns indexed in the SAED indicate that the crystal structure is almost identical to bulk Co<sub>3</sub>O<sub>4</sub>, as further supported by XRD results. Strong reflections of



Fig. 2. Size distribution histogram of the sample from TEM analysis.



Fig. 3. SAED pattern of Co<sub>3</sub>O<sub>4</sub> nanoparticles.

the spinel structure such as (022) and (113) are observed. Trace of second phase is found, this may be due to the residual precursors.

The *d* space values measured from SAED patterns are listed in Table 1 along with the values of bulk  $Co_3O_4$  for reference. The values of both samples are similar to the bulk value. The small deviation of lattice parameters from the bulk value is usually found in nanomaterials. The reflection (1 1 1) can be seen but it partially overlaps to the direct beam. The reflections (2 2 2) and (1 3 3) are not observed in the patterns because they are extremely weak in electron diffraction, even in bulk.

# 3.2. X-ray diffraction studies (XRD)

The XRD pattern of the precursor annealed at 400 °C for 3 h is shown in Fig. 4. The diffraction peaks can be assigned to a cubic phase of  $Co_3O_4$  according to JCPDS 43-1003, indicating the formation of the spinel  $Co_3O_4$ . The small peak intensities reveal the nano-size formation of grains.

# 3.3. Infra-red spectroscopy

The formation of  $Co_3O_4$  phase from the precursor was further ascertained by IR spectroscopy shown in Figs. 5 and 6. In Fig. 5 absorption bands at 2925 and 2856 cm<sup>-1</sup> corresponding to the C–H stretching vibrations and the peak around 1100 and 1030 cm<sup>-1</sup> corresponding to the P=O and P–O stretching and bending vibrations are observed. These are particularly due to the presence of phosphine oxide. In the case of  $Co_3O_4$  nanoparticles (Fig. 6) sharp

Table 1	
Comparison of <i>d</i> space values (Å) for Co <sub>3</sub> O <sub>4</sub> .	

( <i>h k l</i> )	This study	Bulk Co <sub>3</sub> O <sub>4</sub>
(111)	4.69	4.67
(022)	2.86	2.86
(113)	2.44	2.44
(222) <sup>a</sup>	-	2.33
(004)	2.02	2.02
(133) <sup>a</sup>	-	1.85
(224)	1.66	1.65
(333) and (115)	1.56	1.56
(044)	1.43	1.43

<sup>a</sup> The extremely weak reflection in an electron diffraction pattern.



Fig. 4. X-ray diffraction pattern of Co<sub>3</sub>O<sub>4</sub> nanoparticles.

absorptions of Co(III)–O and Co(II)–O stretching vibrations at ~570 and 660 cm<sup>-1</sup> are observed [6]. Very weak absorptions due to P=O and P–O indicate the absorption of phosphate ions on the surface of Co<sub>3</sub>O<sub>4</sub> nanoparticles because they could not be washed completely by acetone.

## 3.4. Thermal analysis

It is known that  $Co_3O_4$  is a thermodynamically stable form under an oxygen containing atmosphere. The stability of the prepared  $Co_3O_4$  was examined under inert gas (nitrogen) atmosphere. The TGA showed a small weight loss (2%) from room temperature to about 160 °C (Fig. 7) which corresponds to loss of structural water and part of the organic decomposition [12]. This is a further evidence of a small amount of surfactant adsorption on the nanoparticles. The TG curve at around 760 °C can be described by the forward reaction of the following equilibrium:

$$Co_3 O_4 = 3CoO + 1/2O_2 \tag{1}$$

Conversion of  $Co_3O_4$  to CoO occurs at >760 °C with a DTG peak centered at 845 °C. This is well within the range of 842–858 °C reported by Z.P. Xu and H.C. Zeng for  $Co_3O_4$  to CoO transfor-



Fig. 6. FTIR spectra of Co<sub>3</sub>O<sub>4</sub> nanoparticles.



**Fig. 7.** TG-DTG thermograms of  $Co_3O_4$  nanoparticles in nitrogen atmosphere in the temperature range from 33–1000 °C and heating rate of 10 °C min<sup>-1</sup>.



**Fig. 8.** Magnetization curve as a function of applied magnetic field of nanoparticulate  $Co_3O_4$  at room temperature. The inset shows the magnification of the hysteresis loop.

mation in hydroxide-derived cobalt compounds under nitrogen atmosphere [17].

#### 3.5. Magnetic measurements

Magnetic measurements performed on synthesized Co<sub>3</sub>O<sub>4</sub> nanoparticles are presented in Fig. 8. From the figure, a low coercive force and remanent magnetization can be seen, which indicated that Co<sub>3</sub>O<sub>4</sub> nanoparticles exhibit a little ferromagnetic property (Fig. 8 inset). The effective magnetic moment per ion  $\mu_{eff}$  for the nanoparticle was estimated to be 4.27  $\mu_{B}$ . This value is slightly higher than that of bulk crystalline Co<sub>3</sub>O<sub>4</sub> (4.14  $\mu_{B}$ ) [18]. Makhlouf [19] has reported the effective magnetic moment per ion to be 4.4  $\mu_{B}$  for Co<sub>3</sub>O<sub>4</sub> of 20 nm size. The slight increase in the magnetic moment and the ferromagnetic behaviour of the nanoparticles can be explained as follows.

Bulk  $Co_3O_4$  has a normal spinel structure with antiferromagnetic exchange between ions occupying tetrahedral and octahedral sites [18]. It has zero net magnetization due to the complete compensation of sub lattice magnetizations. Hence the change from an antiferromagnetic state for bulk  $Co_3O_4$  to a weakly ferromagnetic state for nanoparticulate  $Co_3O_4$  can be ascribed to the uncompensated surface spins and/or finite size effects [13,20,21]. Also the  $Co^{3+}$  ions have no moment at the B sites, while the  $Co^{2+}$  ions at the A sites have a permanent moment of  $3.25 \mu_B$  as found in neutron diffraction experiments [18]. Due to the small size of the particles, it is possible that a noticeable part of  $Co^{3+}$  ( $d^6$  low spin) and  $Co^{2+}$ ( $d^7$  ions) switch octahedral and tetrahedral sites, probably by local electron hopping. As a consequence of such a partly inverted spinel structure, the antiferromagnetic interactions between Cobalt ions on the tetrahedral sites do not annihilate [6].

#### 4. Conclusions

In summary, the cobalt oxide nanoparticles have been successfully synthesized by a quick microwave assisted synthetic route using the surfactant TOPO. The synthesized  $Co_3O_4$  nanoparticles were in the size range of 3–12 nm. FTIR and TG measurements showed a negligible amount of the surfactant on the surface of nanoparticles. A small hysteresis loop was observed at room temperature and the value of the effective magnetic moment  $\mu_{eff}$  was estimated to be 4.27  $\mu_B$ . This could be attributed to the uncompensated surface spins and/or finite size effects. The synthetic route can be further used for the synthesis of other transition metal oxides.

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